



# Magnetic non-uniformity in $(\text{La}_{0.4}\text{Pr}_{0.6})_{0.67}\text{Ca}_{0.33}\text{MnO}_3$ films and measurement of the strain- magnetization coupling coefficient

S. Singh<sup>1,2</sup>, M. R. Fitzsimmons<sup>1</sup>, T. Lookman<sup>1</sup>, H. Jeon<sup>3,4</sup>, M. A. Roldan,<sup>5</sup> M. Varela<sup>3</sup>, and  
A. Biswas<sup>4</sup>

<sup>1</sup>*Los Alamos National Laboratory, Los Alamos, NM 87545, USA*

<sup>2</sup>*Solid State Physics Division, Bhabha Atomic Research Center, Mumbai 400085, India*

<sup>3</sup>*Oak Ridge National Laboratory, Oak Ridge TN 37831 USA*

<sup>4</sup>*Department of Physics, University of Florida, Gainesville, FL 32611, USA*

<sup>5</sup>*GFMC. Dpto. Fisica Aplicada III, Universidad Complutense de Madrid, 28040 Madrid, Spain*

Work supported by the OBES-DOE and the NSF.

# Outline

- Motivation and background
- Sample preparation and characterization
- Possible evidence for phase separation
- Magnetic depth profile  $\sigma = 0$  and consequences
- Magnetic depth profile  $\sigma \neq 0$  and consequences
- Conclusions

## Motivation

- To explore phase separation/co-existence in LPCMO thin films.
- To understand origin of low TMR (attributed to degraded interfacial magnetization).
- To understand the *exclusive* role of strain on magnetism.

Motivation: Clarify the role of stress on ferromagnetism in manganite films, which is decidedly mixed.

Report	Compressive, strengthens FM	Tensile, weakens FM	Compressive, weakens FM	Tensile, strengthens FM
Bulk LCMO & pressure	✓			
Theory 1	✓	✓		
Theory 2			✓	✓
Thickness 1	✓	✓		
Thickness 2			✓	✓
Epi-strain	No effect	✓	No effect	
Chemical pressure	✓	✓		
Phase transformation				✓
Piezoelectric		✓		
Mechanical jigs	$T_{MI}$ increases	$T_{MI}$ decreases		

↑

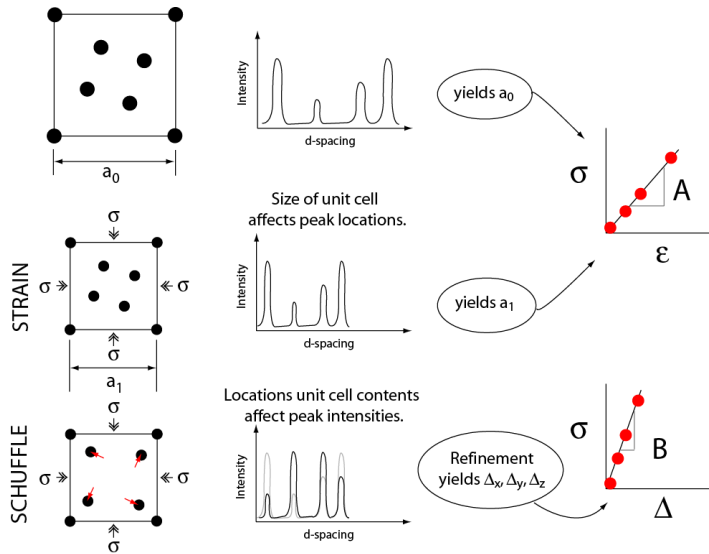
Films

↓

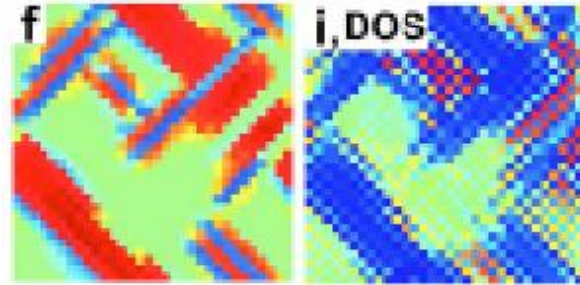
# Background

$$E_{structure} = E_{strain} + E_{atomic} + E_{coupled}$$

$$= A\varepsilon^2 + B\Delta^2 + C\Delta\varepsilon^2$$

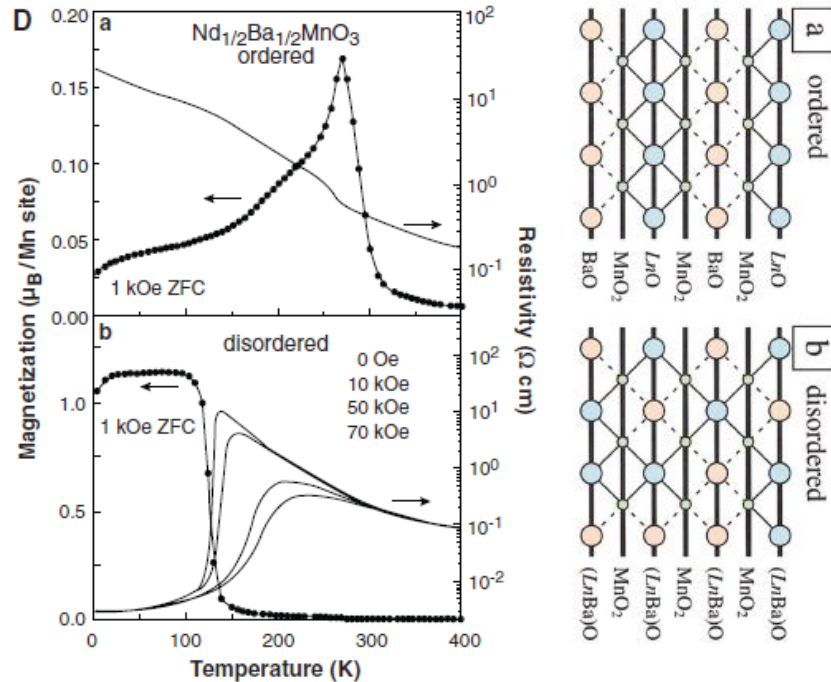


K.H. Ahn et al., Nature **428**, 401 (2004).



- Property sum and product rules are important at the nm scale. R.E. Newnham, D.P. Skinner and L.E. Cross, Mat. Res. Bull. **13**, 525 (1978).
- Also, quenched disorder [e.g., E. Dagotto, Science 309, 257 (2005)].
  - Random fluctuations of dopant density, strain fields, J-T distortions...
- Phase coexistence very sensitive to the environment.

# Complexity in systems that are not “clean”.

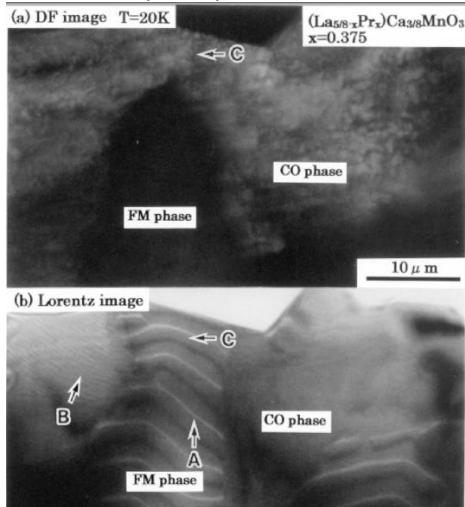


E. Dagotto, Science 309, 257 (2005).

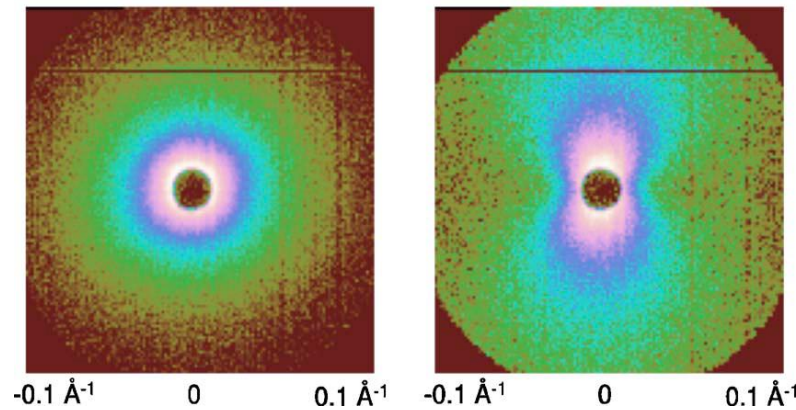
D. Akahoshi et al., PRL 90 1777203 (2003).

# Experimental evidence for phase coexistence.

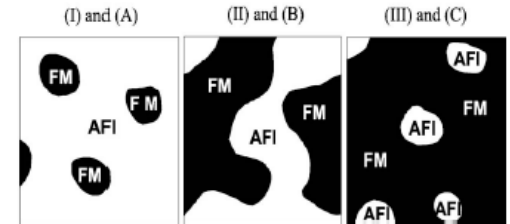
DF and Lorentz images of **bulk** LPCMO. (Mori)



Field dependence SANS data of **bulk** PCMO. (Saurel)

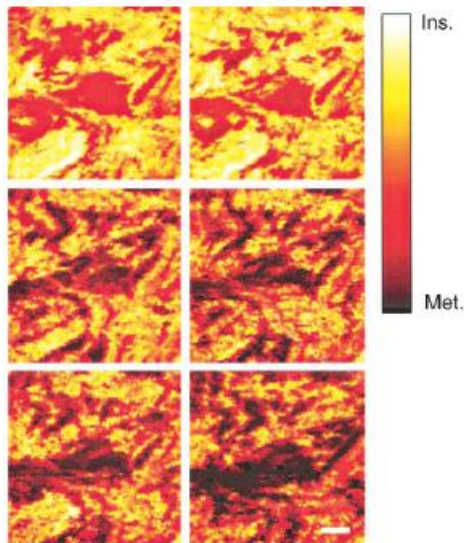


Relevant length scales vary from 100's nm to 10's of microns.

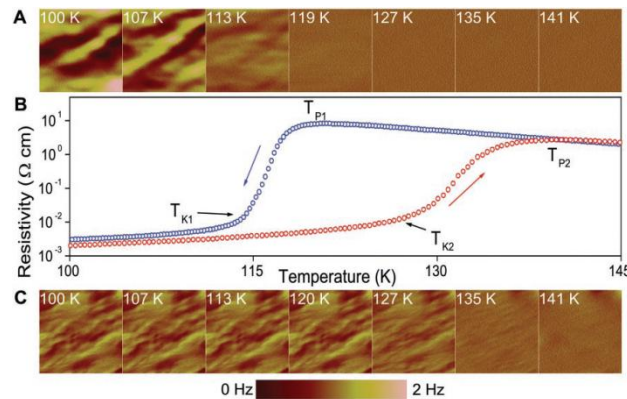


H increases →

STM images of LCMO **thin film** from 0 to 9T. (Fäth)



MFM of LPCMO **thin film**  
Zhang et al., Science 298 805 (2002).

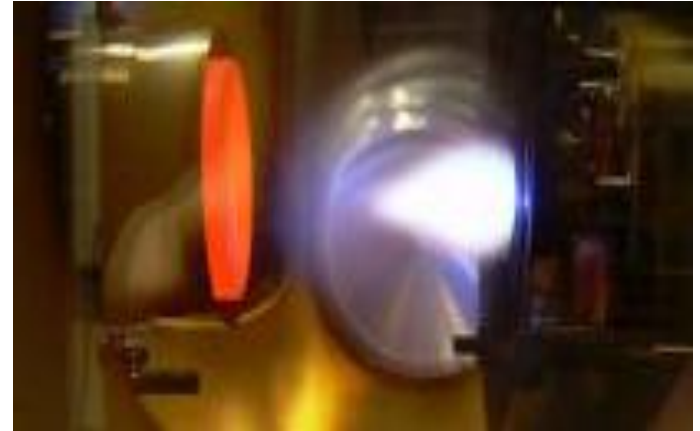


## Compelling evidence

	Electronic phase separation?	Magnetic phase separation?
Bulk	Yes	Yes
Film	Yes	?

# Sample preparation

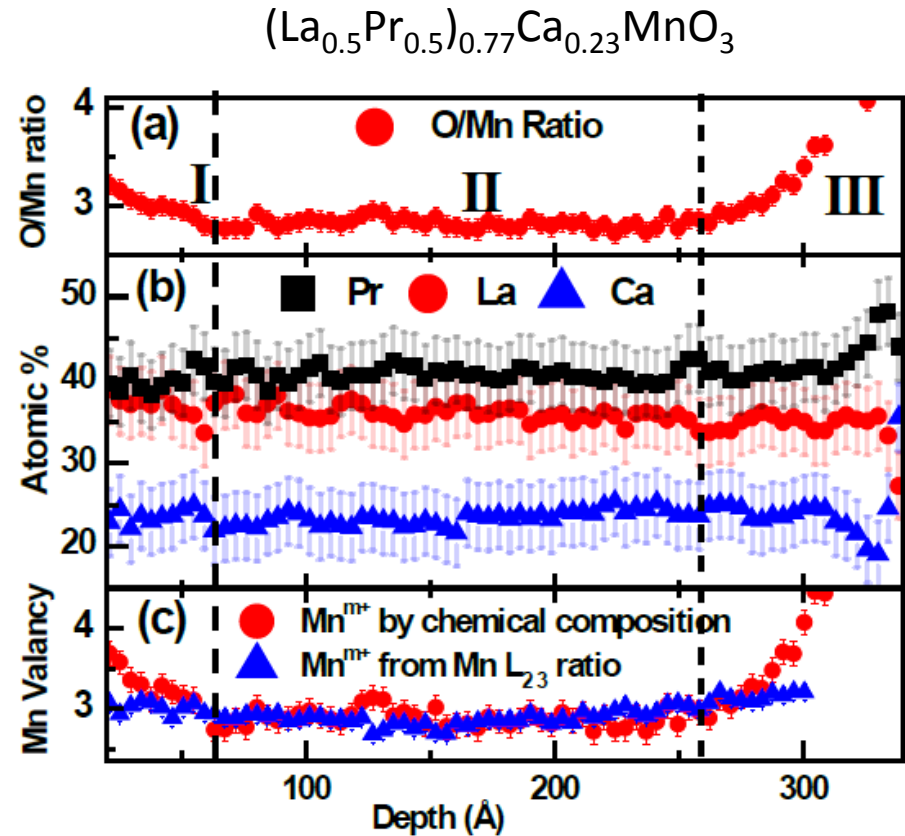
- Samples grown by PLD (A. Biswas, UFL).
- Target composition:  
 $(\text{La}_{0.4}\text{Pr}_{0.6})_{0.67}\text{Ca}_{0.33}\text{MnO}_3$
- (110)  $\text{NdGaO}_3$  (NGO) substrates are 1cm by 1cm by 250 $\mu\text{m}$ .
- 30 nm thick (101) LPCMO single crystal films.
- Small epi-strain:  
+0.4% || [001] NGO  
+0.2% || [-110] NGO  
relative to bulk LPCMO.





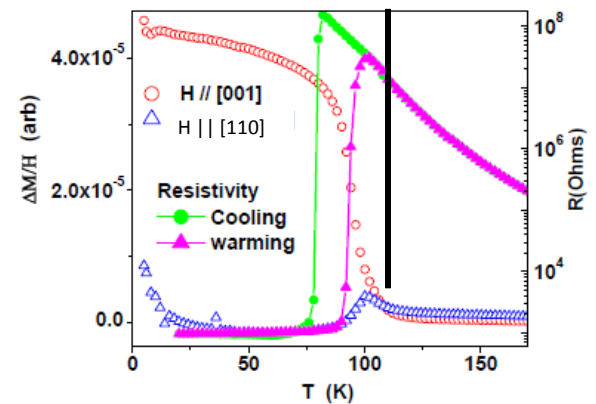
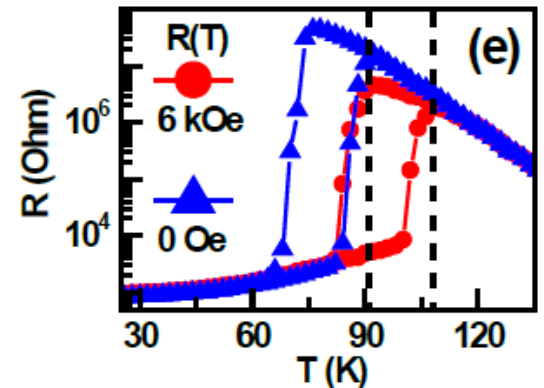
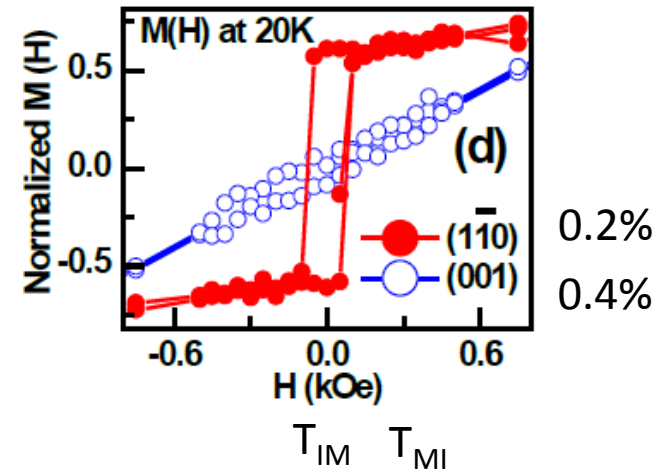
# Electron energy loss spectroscopy

- Chemically nonuniform.
- Excess  $\text{Mn}^{4+}$  at surface and buried interface.
- Excess  $\text{Mn}^{4+}$  due to excess O, not Ca deficiency.



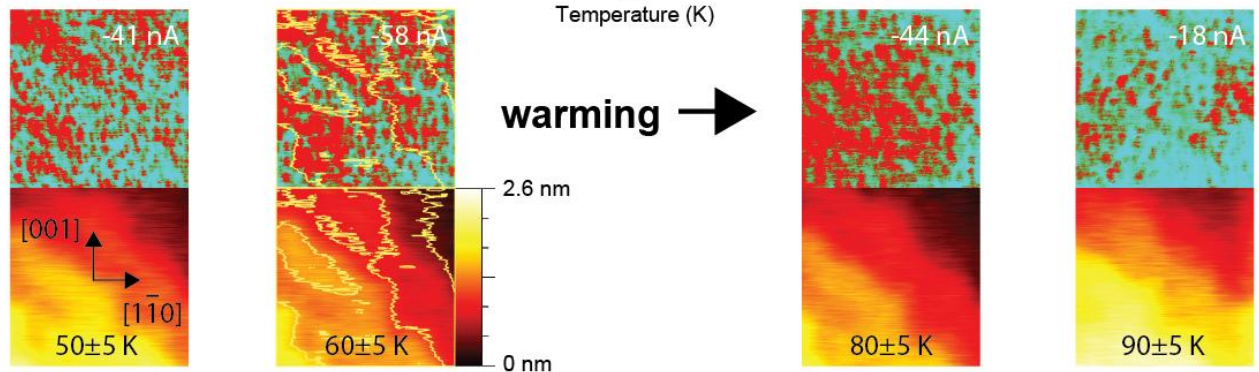
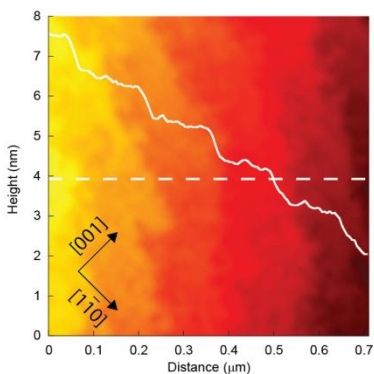
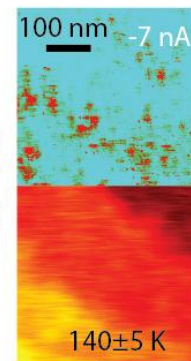
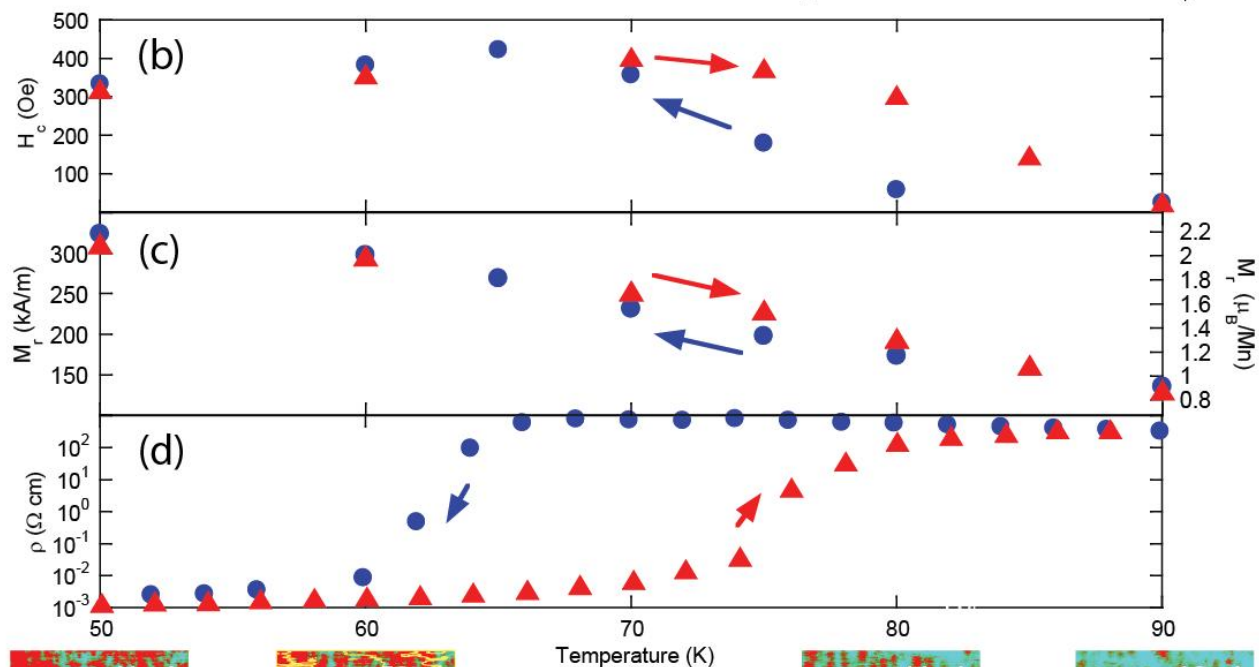
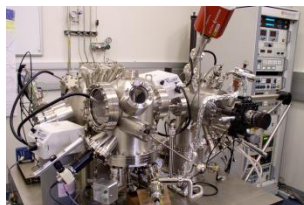
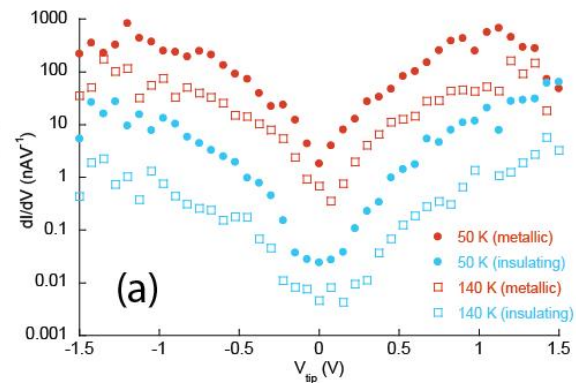
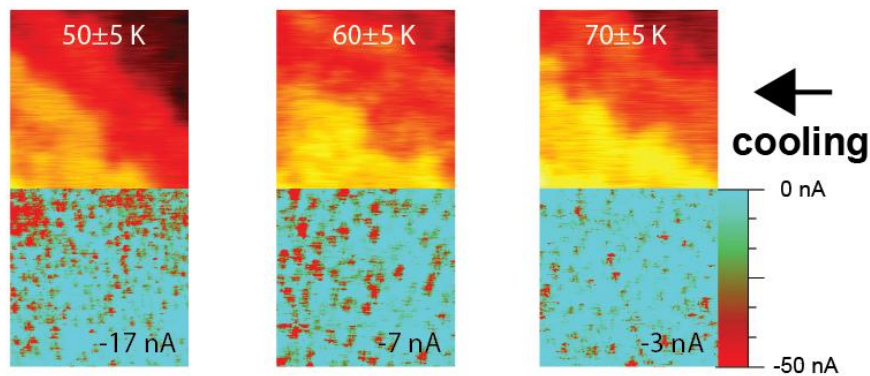
# Magnetometry

- Strong in-plane anisotropy.
- Field favors metallic phase.
- Metal-insulator transitions are not the same as the Curie temperature.



Performed at the ANL CINT facility (w/ J. Guest).

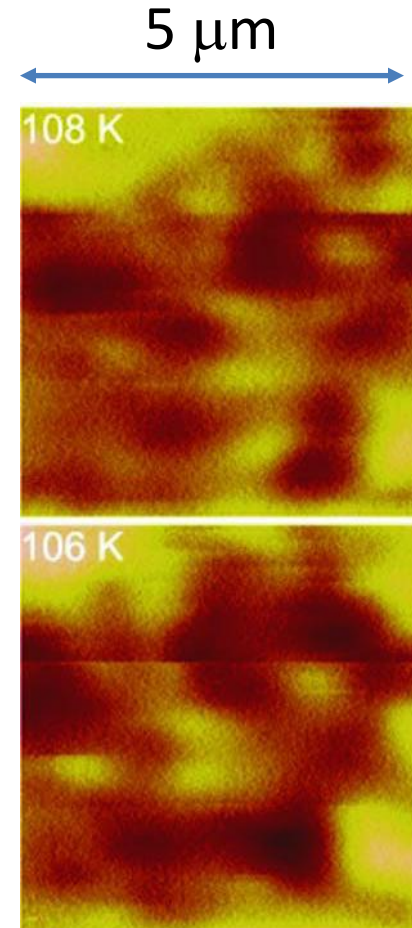
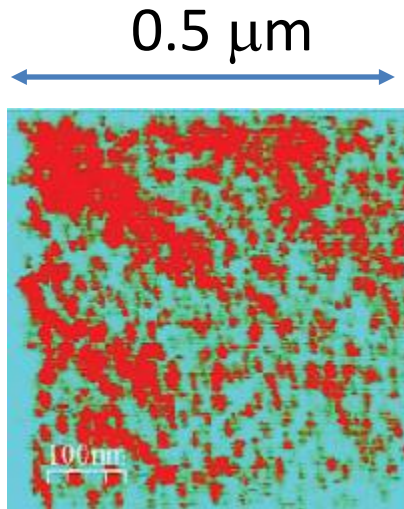
Temperature changed at 0.4K/min.



The length scales for electronic and magnetic texture do not match for films of the same nominal composition.

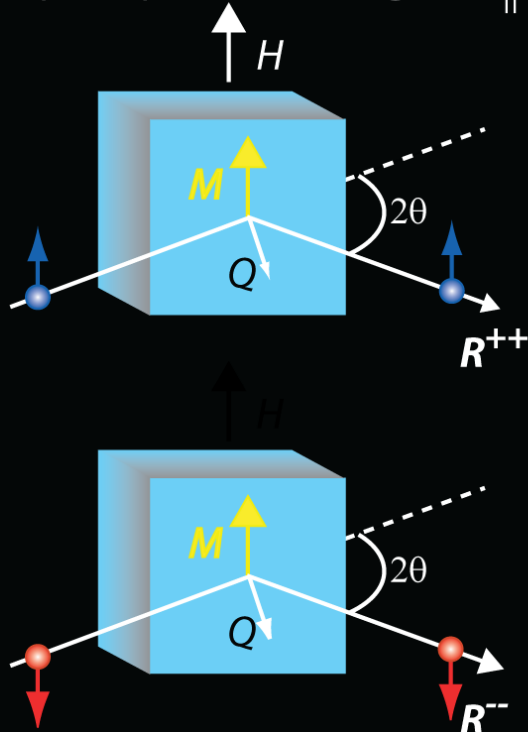
**cAFM (this work)**

**MFM (Lozanne, Phys. Today 1/3)**

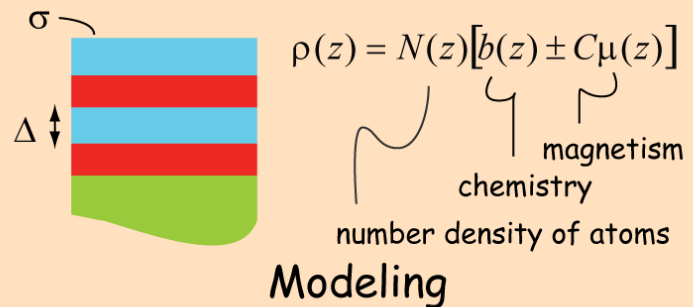
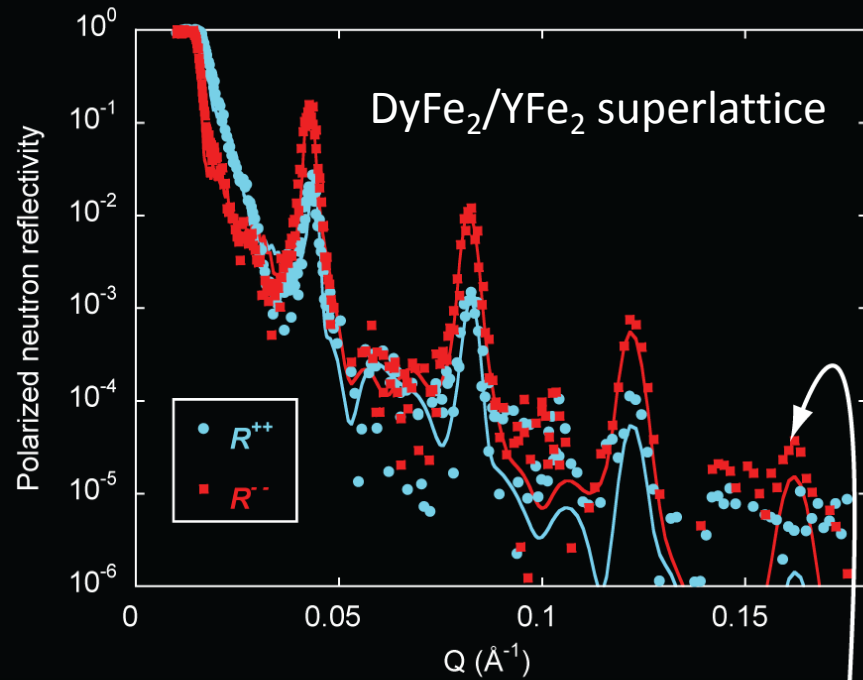


# Polarized Neutron Reflectometry

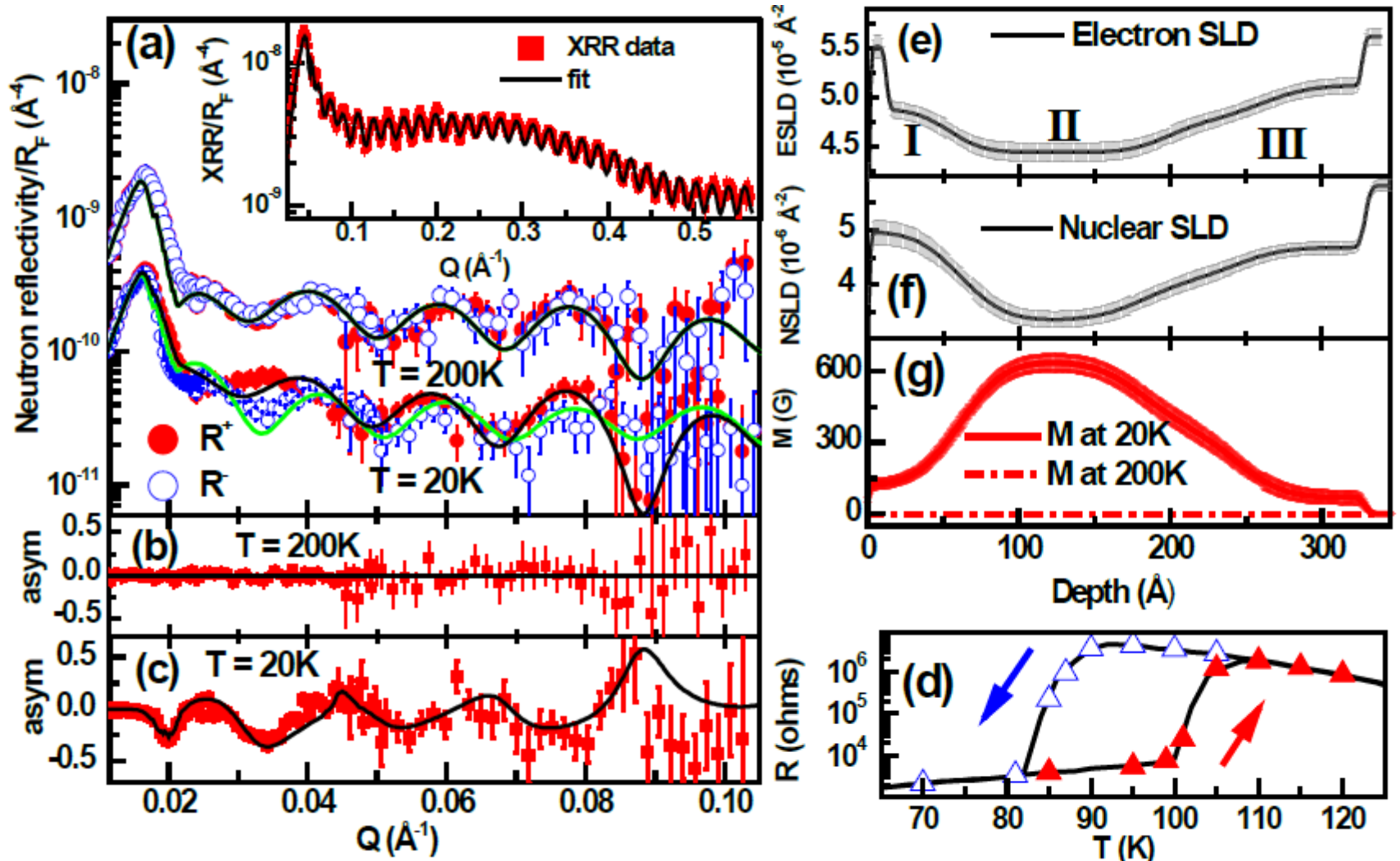
non-spin-flip reflectivities give  $\overline{M_{\parallel}}(Q)$



Measurement of the neutron scattering as a function of wavevector transfer (of order  $0.1 \text{ \AA}^{-1}$ ) allows us to determine the structures and properties of materials that are non-uniform with nanometer resolution.



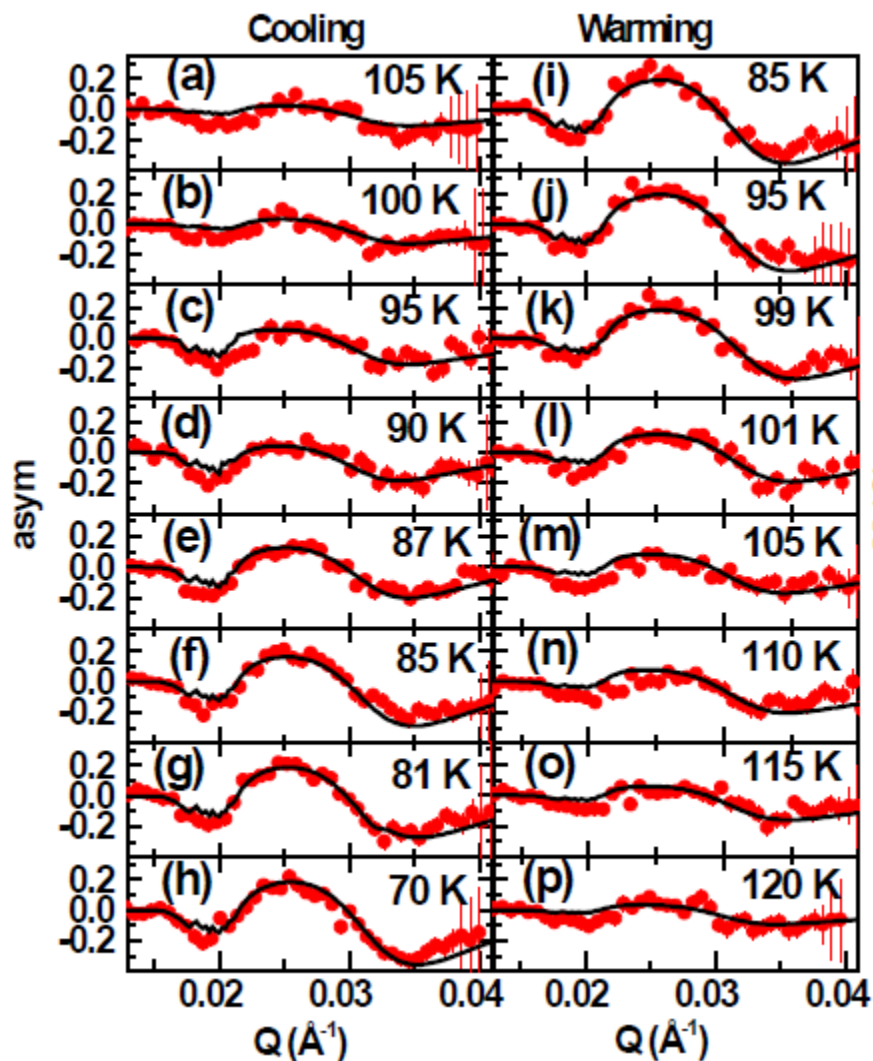
Chemical and magnetic depth profiles are non-uniform ( $\sigma = 0$ )



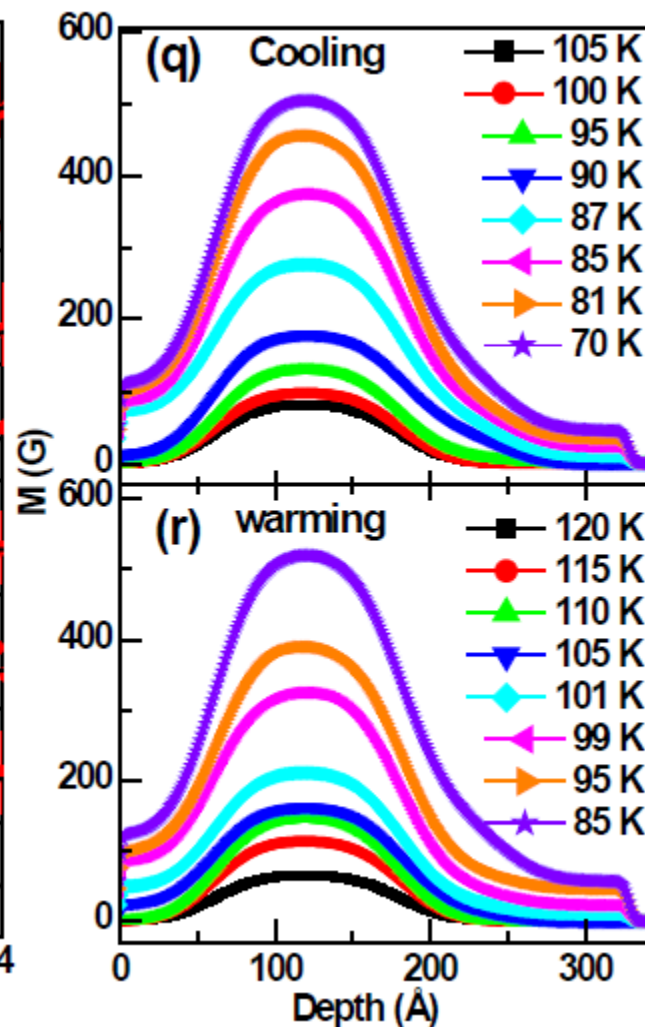
$H = 6 \text{ kOe}$

# Temperature dependence of the saturation magnetization

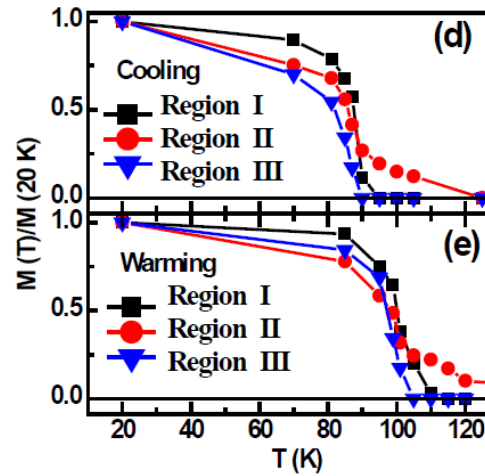
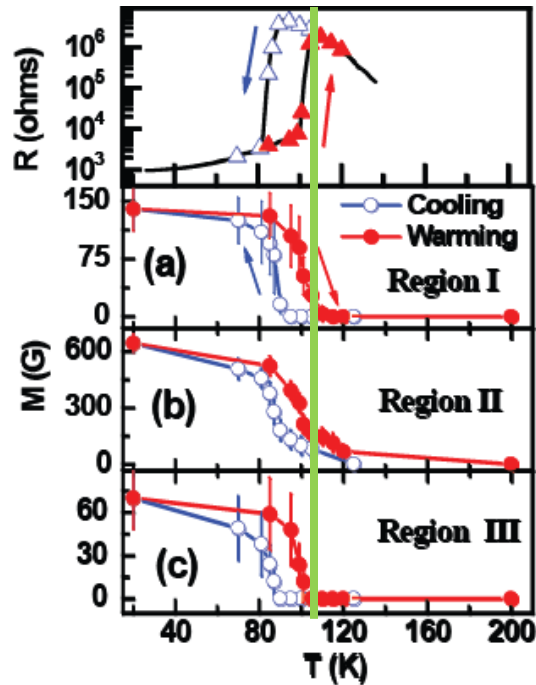
Spin asymmetry of the neutron reflectivity



Magnetization depth profiles



Saturation magnetization is less in Mn<sup>4+</sup> rich regions.



$H = 6 \text{ kOe}$

(1)  $M$  shows hysteresis similar to  $R$ .

(2)  $T_c$  of region II exceeds  $T_{MI}$

(3)  $M_s$  suppressed in Mn<sup>4+</sup> rich regions (more AF interactions?).

S. Singh et al., PRL **108**, 077207 (2012)

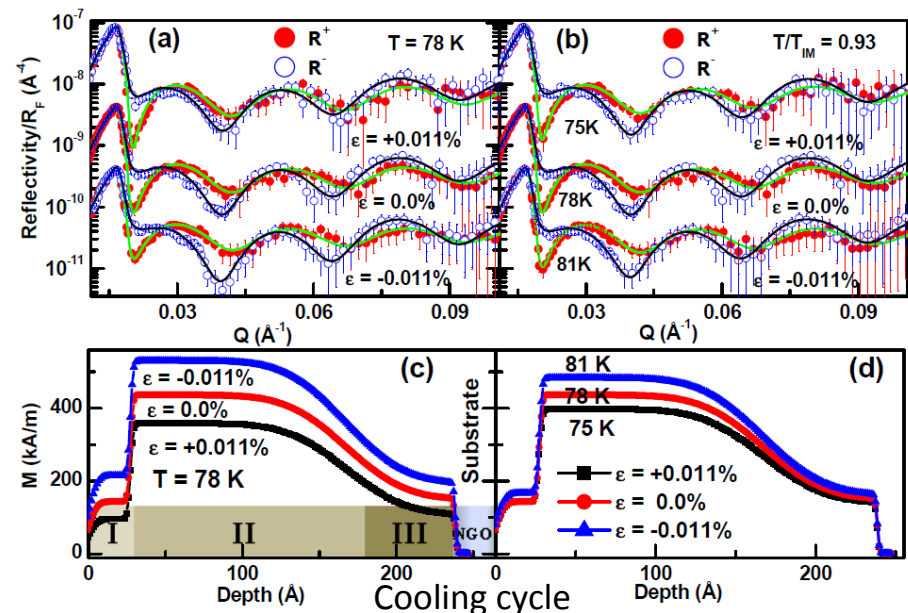
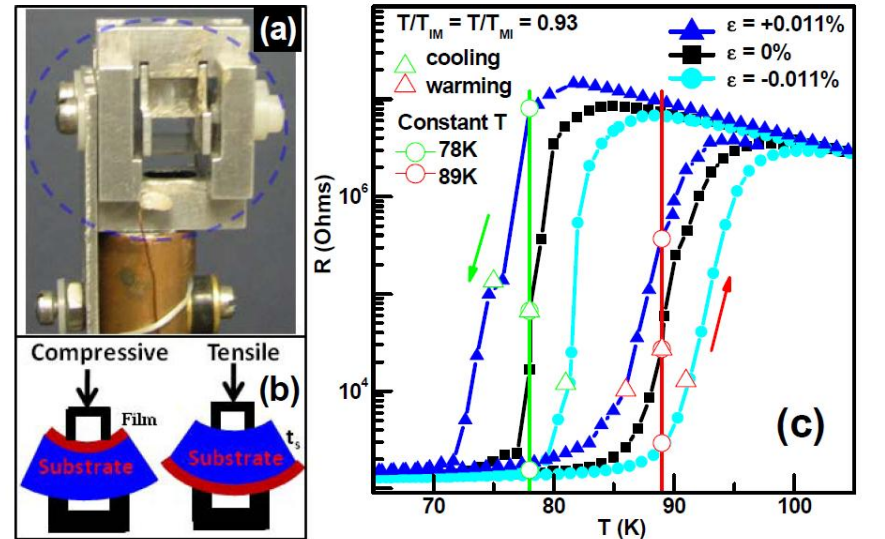


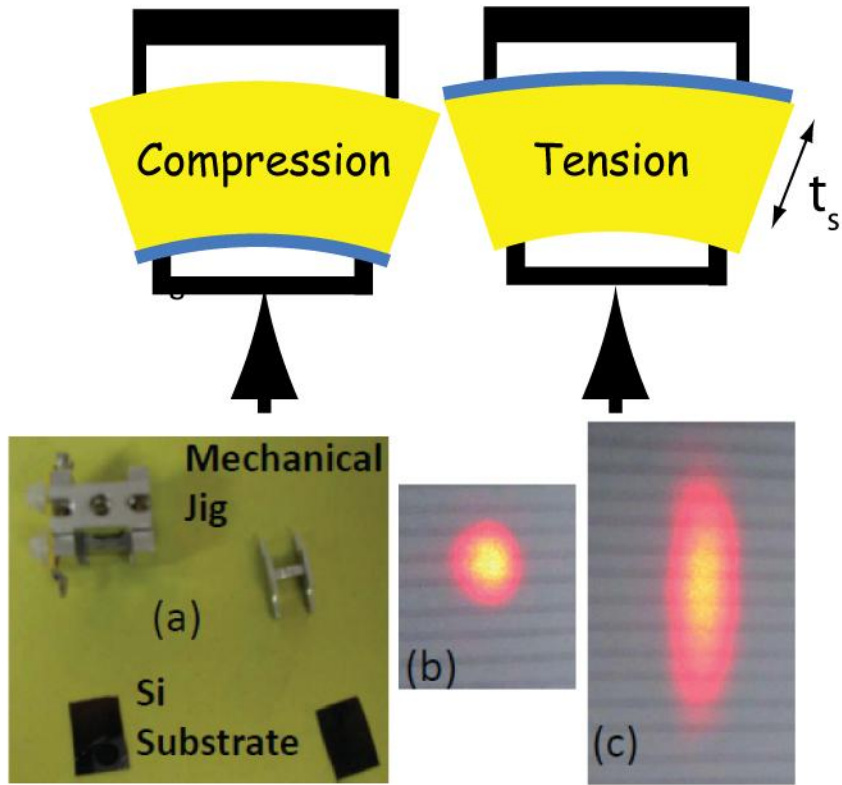
Compressive stress stabilizes  $T_{IM}$  and  $T_{MI}$  to higher T and increases  $M_s$ .

- 4 point mechanical jig produces  $\varepsilon = \pm 0.011\%$ .
- Neutron scattering and transport measured vs.  $\sigma$ , H and T.
- Collected data for constant T and constant  $T/T_{IM,MI}$ .
- Compressive stress ( $-\varepsilon$ ) increases  $M_s$  ( $T_{MI}$  &  $T_{IM}$ ).

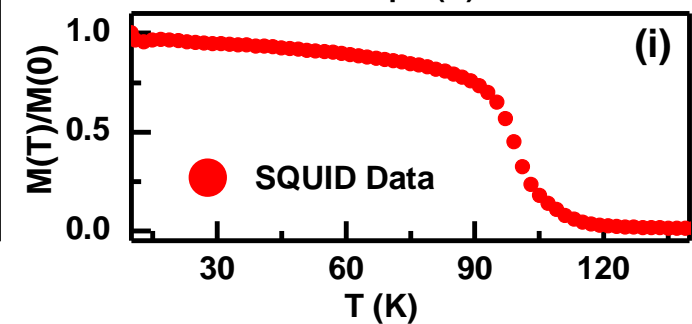
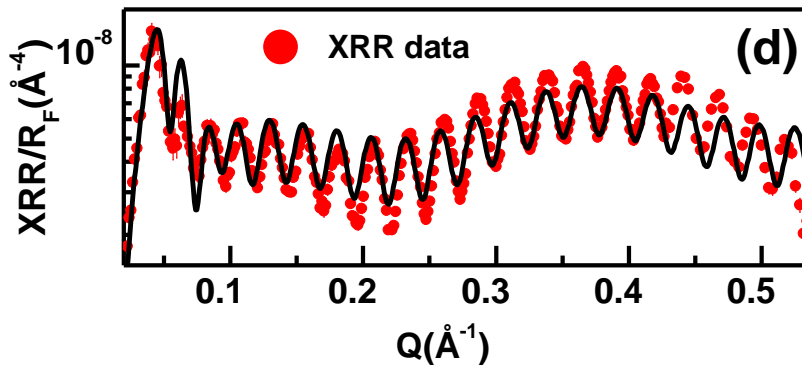
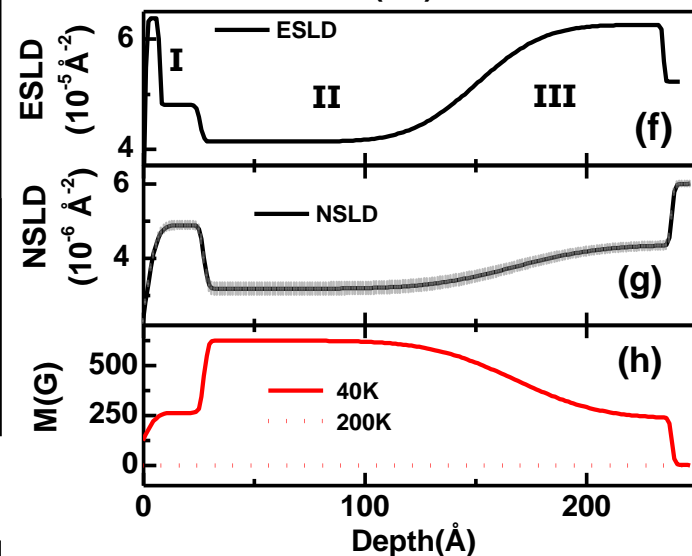
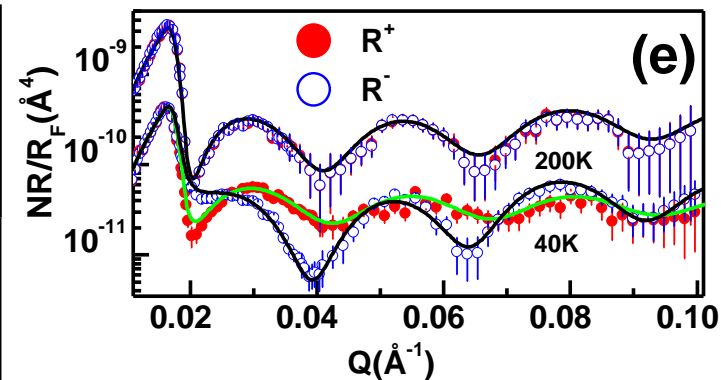
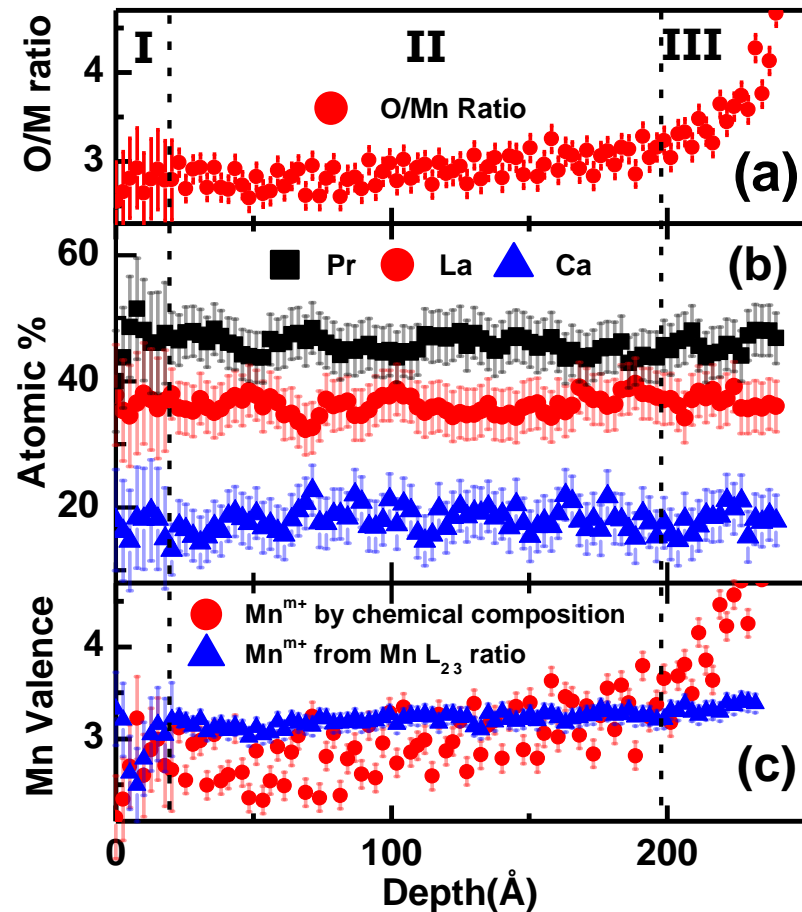
H = 6 kOe

<http://arxiv.org/abs/1201.4001>



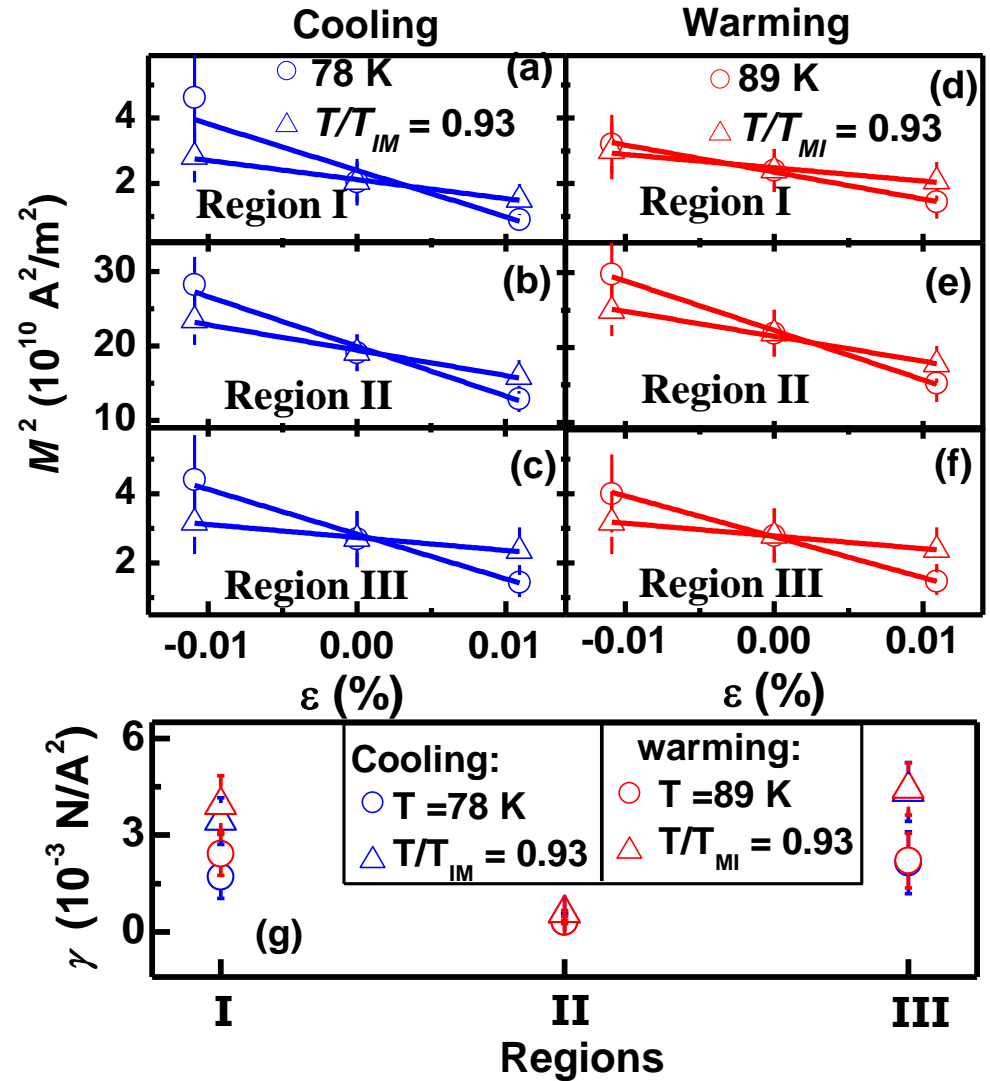


# Characterization of the film: EELS, XRR, PNR



# Magnetic depth profile & $\sigma \neq 0$

- $F_C = \gamma \epsilon M_\epsilon^2 + \frac{A}{2} \epsilon^2$
- $M^2 = M_0^2 - \frac{A}{\gamma} \epsilon$
- $A = 200$  GPa
- $\gamma$  does not depend upon cooling or warming cycles.
- $\gamma$  smallest for film bulk (least  $Mn^{4+}$ ) implies strongest coupling.
- $\gamma \sim 0.0003-0.0006$  N/A<sup>2</sup>



# Conclusions

- Length scales of electronic and magnetic texture may differ.
- Length scale of electronic texture confined by terrace steps.
- The LPCMO films are neither chemically nor magnetically uniform with depth.
- Compressive *elastic* strain (-'ve  $\varepsilon$ )
  - Increases  $M_s$ .
  - Favors the ferromagnetic phase.
  - Increases the metal-insulator transition temperatures.
- Coupling between strain and ferromagnetism is strongest for the bulk film composition (i.e., not  $\text{Mn}^{4+}$  rich).
- Demonstrated several technical innovations that can be broadly applied to other systems, especially multiferroic, and piezomagnetic films.